

# Modification of TiO<sub>2</sub> Electrode with Organic Silane Interposed Layer for High-Performance of Dye-Sensitized Solar Cells

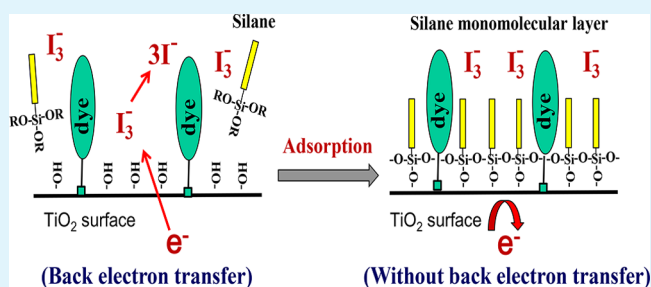
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## S Supporting Information

**ABSTRACT:** Back electron transfer from the TiO<sub>2</sub> electrode surface to the electrolyte is the main reason behind the low open circuit potential ( $V_{oc}$ ) and the low fill factor (FF) of the dye-sensitized solar cells (DSSCs). Modifications to the TiO<sub>2</sub> electrode, fabricated using {010}-faceted TiO<sub>2</sub> nanoparticles with six different kinds of silane, are reported to decrease the back electron transfer on the TiO<sub>2</sub> surface. The effect of alkyl chain length of hydrocarbon silanes and fluorocarbon silanes on adsorption parameters of surface coverage and adsorption constant, interfacial resistance, and photovoltaic performances were investigated. Adsorption isotherms, impedance analysis, and photovoltaic measurements were used as the investigation techniques. The reduction of back electron transfer depended on the TiO<sub>2</sub> surface coverage by silane, alkyl chain length, and the molecular structure of the silane. Even though  $V_{oc}$  and FF were improved, significant reduction in short-circuit photocurrent density ( $J_{sc}$ ) was observed after silanization because of desorption of dye during silanization. A new approach, sequential adsorption process of silane and dye, was introduced to enhance  $V_{oc}$  and FF without lowering  $J_{sc}$ . Heptadecafluorodecyl trimethoxy-silane showed the highest coverage on the surface of the TiO<sub>2</sub> and had the highest effect on the performance improvement of the DSSC, where  $V_{oc}$ , FF, and efficiency ( $\eta$ ) were improved by 22, 8.0, and 22%, respectively.

**KEYWORDS:** dye-sensitized solar cells, adsorption isotherms, silanization, back electron transfer, photovoltaic performances, sequential adsorption



## 1. INTRODUCTION

Dye-sensitized solar cells (DSSCs) have attracted great interest from scientists because they have the potential to convert sunlight (solar power) to electricity (electrical power) at a low cost in an environmentally friendly way.<sup>1</sup> A typical DSSC consists of a dye-sensitized TiO<sub>2</sub> electrode, a Pt electrode, and an electrolyte. Figure 1 shows the working principle of the DSSC; namely, (1) excitation process, (2) injection process, (3) energy generation, (4) regeneration of the mediator, and (5) regeneration of dye.<sup>2</sup> However, as shown in Figure 1, it is possible for backward reactions to occur, which would yield a substantially diminished return. The possible backward reactions are (6) the excited electron in LUMO (lowest unoccupied molecular orbital) level of the dye can be recombined with the hole in its HOMO (highest occupied molecular orbital) level, (7) the injected electrons in the TiO<sub>2</sub> conduction band can be recombined with I<sub>3</sub><sup>-</sup> ions in the electrolyte, (8) the electrons in the TiO<sub>2</sub> conduction band can be recombined with holes in the HOMO level of the dye, and (9) the electrons in the conducting glass can be recombined with I<sub>3</sub><sup>-</sup> ions in the electrolyte.<sup>3</sup>

These backward reactions will result in decreased efficiencies of DSSCs. But dye regeneration reaction (5) and electron

injection from the LUMO level of the dye to the conduction band of TiO<sub>2</sub> (2) are taken at fast rates (ns and ps).<sup>4</sup> The length of time takes for an electron to diffuse through TiO<sub>2</sub> nanoparticles to the conducting glass surface is about 10 ms.<sup>3,4</sup> This provides ample time for the back electron transfer from TiO<sub>2</sub> surface to oxidized redox species in the electrolyte. Therefore, a major limitation of DSSCs efficiencies is the loss of injected electrons from the semiconductor to the oxidized redox species in the electrolyte. That means the reaction shown by arrow (7) in Figure 1 is the prominent backward reaction that should be eliminated to get high efficiency. Our recent study on dye adsorption on TiO<sub>2</sub> electrode surfaces has suggested that only about one-third of the TiO<sub>2</sub> electrode surface is covered by the adsorbed-dye molecules and two-thirds of the surface is exposed to the electrolyte solution.<sup>5</sup> The uncovered area is potential site for interfacial back electron transfer which is represented by arrow (7). Thus, it is necessary to reduce the rate of the backward reaction at the TiO<sub>2</sub> surfaces in DSSCs.

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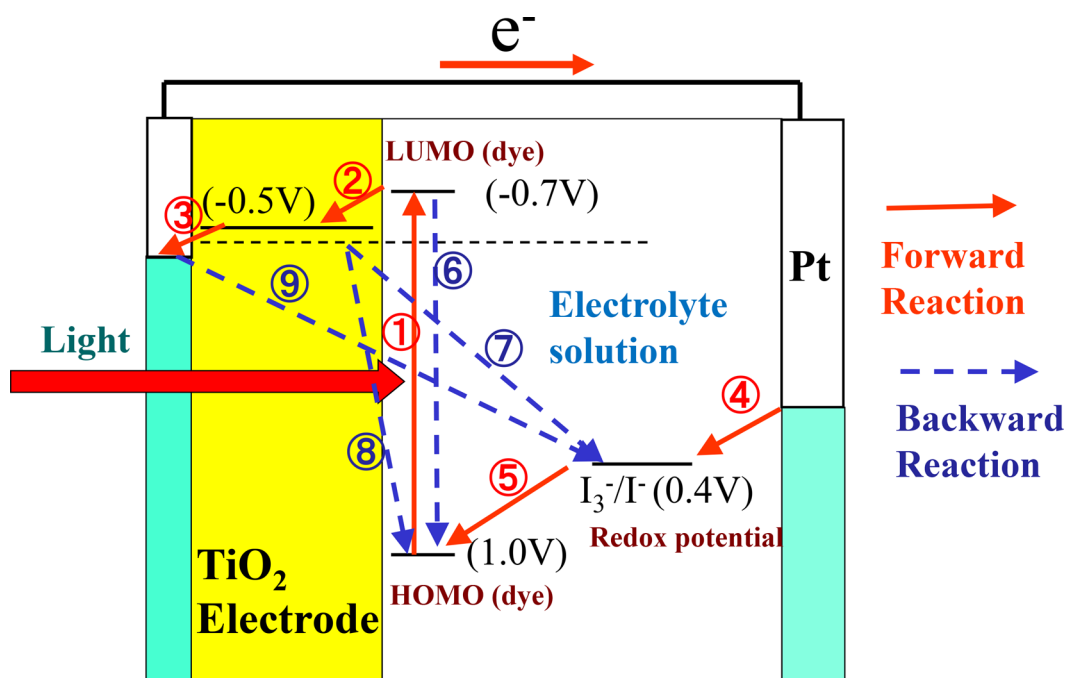


Figure 1. Working principle of a typical dye-sensitized solar cell.

Extensive studies on minimizing the back electron transfer have been reported. The addition of additives to redox electrolyte, such as 4-*tert*-butylpyridine (TBP) and guanidinium thiocyanate (GuSCN), have suppressed the back electron transfer by adsorbing these additives on the surface of the TiO<sub>2</sub> electrode.<sup>6–9</sup> Because these cations formed weak bonds on the TiO<sub>2</sub> surface, stronger surface modification technique should be investigated. The TiO<sub>2</sub> under layer on conducting glass substrate (FTO) has reduced the back electron transfer on the FTO surface.<sup>10,11</sup> But the back electron transfer on the TiO<sub>2</sub> surface is much higher than that on the FTO surface because of the large surface area of TiO<sub>2</sub>, that is two-thirds of its surface, is exposed to the electrolyte solution.<sup>5</sup> Limited studies have been reported in TiO<sub>2</sub> surface passivation using silanes.<sup>3,12–14</sup> The silane adsorption process is a chemisorption process. Therefore, it can cover the exposed TiO<sub>2</sub> surface with a covalently bonded strong silane monomolecular layer to decrease the rate of the back electron transfer by physically preventing the close approach of I<sub>3</sub><sup>−</sup> to the TiO<sub>2</sub> electron surface.

A couple of studies have reported on the dye-sensitized TiO<sub>2</sub> surface passivation,<sup>3,12</sup> using CH<sub>3</sub>SiCl<sub>3</sub> as a silanization agent with the kinetically fast redox couples. Another couple of approaches have been reported to reduce the back reaction using alkyl(trialkoxysilanes with iodine redox couple, and it suggested that the DSSCs performance depends on the length of the alkyl chain of the silane and the composition of the electrolyte.<sup>13,14</sup> Nevertheless, although silane adsorption is an effective way to enhance DSSC performance, to the best of our knowledge, it has not been reported so far on the effects of silanes adsorption parameters of silane coverage and adsorption constant, concentration, and the treatment sequences of silanes on the photovoltaic performances.

In this study, we describe the effects of alkyl-chain length and substitution of the silane coupling agent on the adsorption parameters of adsorption density and adsorption constant, interfacial resistance, electron lifetime, and photovoltaic

performances of DSSCs. We focus on the modification of the exposed TiO<sub>2</sub> surface because in our previous study, we found that anatase cubic {010}-faceted TiO<sub>2</sub> nanoparticles show a specific high short-circuit current density ( $J_{sc} = 21 \text{ mA/cm}^2$ ), but low open-circuit voltage ( $V_{oc}$ ) and fill factor (FF) as a result of the low dye coverage (33%) on the TiO<sub>2</sub> surface, which leads a large back electron transfer.<sup>5</sup> It suggests that a high-performance DSSC can be obtained using {010}-faceted TiO<sub>2</sub> nanoparticles by blocking the back electron transfer. In-depth studies on silanization processes are carried out to minimize the dye desorption during silanization and to optimize the DSSC performance by optimizing the silanization process.

## 2. EXPERIMENTAL SECTION

**2.1. Chemicals and Reagents.** N719 (cis-di (thiocyanate)bis-(2,2'-bipyridyl-4,4'-dicarboxylate)-ruthium(II) (bis-tetrabutylammonium) purchased from Sigma-Aldrich, was used as the dye sensitizer. Other chemicals and reagents were analytical grade and used as received. Six kinds of siloxanes with different end groups as given in Table 1 were used in the present study. These siloxanes were purchased from Shin-Etsu Chemicals.

**2.2. Silane Adsorption on TiO<sub>2</sub> Nanoparticles.** A 50 mg portion of TiO<sub>2</sub> nanoparticles were added to 10 mL of the solution of

Table 1. Surface Modifying Organic Silane Agents Used in This Study

chemical name	chemical formula	abbreviation
trimethoxy(3,3,3-trifluoropropyl)-silane	CF <sub>3</sub> C <sub>2</sub> H <sub>4</sub> Si(OCH <sub>3</sub> ) <sub>3</sub>	FOS-C3F3
hexyltrimethoxy-silane	C <sub>6</sub> H <sub>13</sub> Si(OCH <sub>3</sub> ) <sub>3</sub>	HOS-C6
decyltrimethoxy-silane	C <sub>10</sub> H <sub>21</sub> Si(OCH <sub>3</sub> ) <sub>3</sub>	HOS-C10
octadecyltrimethoxy-silane	C <sub>18</sub> H <sub>37</sub> Si(OCH <sub>3</sub> ) <sub>3</sub>	HOS-C18
heptadecafluorodecyl trimethoxy-silane	C <sub>10</sub> F <sub>17</sub> H <sub>4</sub> Si(OCH <sub>3</sub> ) <sub>3</sub>	FOS-C10F17
N-[3-(trimethoxysilyl)propyl] aniline-silane	C <sub>6</sub> H <sub>5</sub> N(CH <sub>2</sub> ) <sub>3</sub> Si(OCH <sub>3</sub> ) <sub>3</sub>	KMB

silane in ethanol and the silane concentration was maintained within a range of 0.01–0.3 mol/L. Then, it was allowed to adsorb at room temperature for the desired length of time. After that, the solution was centrifuged at a rate of 4000 rpm for 10 min and the supernatant was removed. Solid phase was washed in ethanol to remove physically adsorbed silanes. Then, it was dried at 60 °C for 24 h followed by TG-DTA analysis. The adsorbed amount of silane was measured using weight loss in a temperature range of 200–500 °C in TG-DTA curves, (see Figure S1 in the Supporting Information) which correspond to the decomposition of silane.<sup>15</sup>

**2.3. Fabrication of TiO<sub>2</sub> Electrode.** Anatase {010}-faceted TiO<sub>2</sub> nanoparticles with cubic morphology, particle size of about 20 nm, and BET surface area of 131 m<sup>2</sup>/g, were used in the fabrication of TiO<sub>2</sub> electrode. The TiO<sub>2</sub> nanoparticles were synthesized using the hydrothermal soft chemical method from layered titanate nanosheets.<sup>16</sup> The TiO<sub>2</sub> paste and the TiO<sub>2</sub> electrode were prepared as described in our previous study.<sup>5</sup> The thickness of the TiO<sub>2</sub> porous film was controlled in a range of 8–13 μm. For the fabrication of an optimized TiO<sub>2</sub> electrode, leaflike anatase particles<sup>16</sup> with a size of 150 nm in length and 30 nm in width were used in the scattering layer.

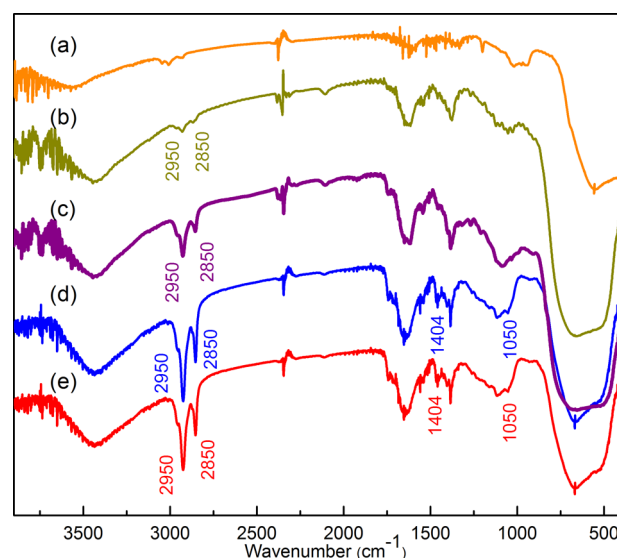
**2.4. Dye and Silane Adsorptions on TiO<sub>2</sub> Electrode.** A 0.3 mmol/L N719 dye solution was used for dye adsorption. The dye loaded TiO<sub>2</sub> electrode was immersed in solution of silane in ethanol for 2 h. The electrode was then rinsed copiously in ethanol and allowed to dry at room temperature.

**2.5. Fabrication of DSSC.** The DSSC was composed of the dye-adsorbed TiO<sub>2</sub> electrode and a Pt-coated FTO conducting glass counter-electrode, with an electrolyte solution between the electrodes. Two types of electrolytes were employed in this work: (1) 0.6 M LiI and 0.03 M I<sub>2</sub> in a mixed solvent of acetonitrile and valeronitrile (85%:15% volume ratio); (2) 0.6 M butylmethylimidazolium iodide, 0.01 M I<sub>2</sub>, 0.1 M LiI, 0.1 M guanidinium thiocyanate, and 0.4 M 4-tert-butylpyridine in a mixed solvent of acetonitrile and valeronitrile (85%:15% volume ratio).

**2.6. Characterization.** Fourier transform infrared (FT-IR) spectra were obtained using a PerkinElmer Spectrum One spectrometer at a resolution of 2 cm<sup>-1</sup> using the KBr technique in the range of 300 to 4000 cm<sup>-1</sup>. The photocurrent–voltage (*I*–*V*) curves were obtained using a Hokuto-Denko BAS100B electrochemical analyzer with a YSS-E40 Yamashita Denso solar simulator (AM 1.5; 100 mW/cm<sup>2</sup>). Electrochemical impedance measurements were performed with an impedance analyzer (Solartron SI 1287) under dark condition. The impedance spectra were recorded in a frequency range of 0.1 Hz to 1 MHz with alternate current (AC) amplitude of 10 mV at an applied direct current (DC) bias of –0.7 V. Thermogravimetric and differential thermal analyses (TG-DTA) were carried out using a Shimadzu DTG-60H at a heating rate of 10 °C/min.

### 3. RESULTS AND DISCUSSION

**3.1. Adsorption of Organic Silanes on TiO<sub>2</sub> Nanoparticles.** The IR spectra of TiO<sub>2</sub> powder samples after the FOS-C3F3 silane adsorption treatment at different time intervals are shown in Figure 2. In the IR spectra, absorption bands at 2850 and 2950 cm<sup>-1</sup> can be assigned to the asymmetric and symmetric stretching of CH<sub>2</sub> groups on the alkyl chain, absorption band at 1404 cm<sup>-1</sup> to the stretching of C–F bond, and absorption band at 1050 cm<sup>-1</sup> to stretching of Si–O, respectively.<sup>14,17–19</sup> The intensity ratios of the silane (Si–O) vibration band to Ti–O vibration band of TiO<sub>2</sub> at 700 cm<sup>-1</sup> increased with increasing the adsorption treatment time up to 2 h, and then became almost constant (see Figure S2 in the Supporting Information). These results reveal that FOS-C3F3 silane molecules are adsorbed on the TiO<sub>2</sub> particle surface, and the silane adsorption almost reaches equilibrium after 2 h. Therefore, a silanizing treatment time of 2 h was used in the next steps.



**Figure 2.** IR spectra of TiO<sub>2</sub> electrodes silanized with FOS-C3F3 for (a) 0, (b) 0.5, (c) 1, (d) 2, and (e) 5 h.

To investigate the effect of silane concentration, alkyl chain length, and substitutes on the self-assembled monolayers (SAMs) formation, adsorption isotherm studies were performed with six different modifiers (Table 1) with the TiO<sub>2</sub> nanoparticles at room temperature. Adsorption isotherm results are shown in Figure 3. To more clearly understand the situation of the silane molecules adsorbed on the TiO<sub>2</sub> nanoparticle surface, we present the silane adsorption amount as per BET specific surface area (*S*<sub>BET</sub>) of the powder sample that corresponds to silane adsorption density on the nanoparticle surface (mol/m<sup>2</sup>). The experimental data of the silanes adsorption were compliant with the Langmuir isotherm, indicating Langmuir monolayer adsorption, meaning typical chemisorption, on the TiO<sub>2</sub> nanocrystals.<sup>5,6,20</sup>

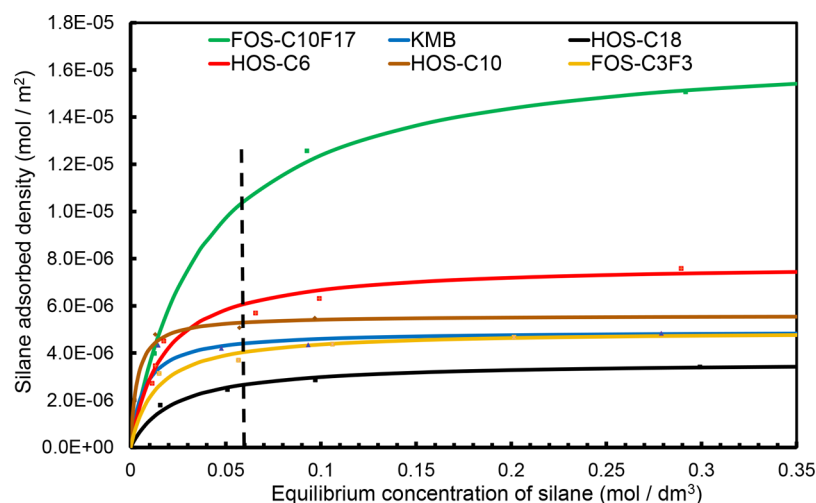
The Langmuir equation can be represented by the following linear formula:

$$C/Q = 1/(Q_m K_{ad}) + C/Q_m \quad (1)$$

where *C*, *Q*, *Q<sub>m</sub>*, and *K<sub>ad</sub>* are the equilibrium concentration of the silane solution, adsorption density at concentration *C*, saturation or maximum adsorption density, and adsorption constant, respectively. The least-squares values (*R*<sup>2</sup>) of the experiment results fitting to formula 1 are 0.997 for HOS-C3F3, 0.997 for HOS-C6, 0.997 for HOS-C10, 0.999 for HOS-C18, 0.999 for FOS-C10F17, and 0.999 for KMB. The adsorption constant and maximum adsorption density can be calculated by plotting *C/Q* against *C*, and the results are tabulated in Table 2. The silanes adsorption densities (*Q*<sub>0.06</sub>) at 0.06 M silane concentration are also given in Table 2 because this concentration of silane solutions is used in the TiO<sub>2</sub> electrode modification. The TiO<sub>2</sub> surface coverage by adsorbed silane molecules can be calculated using following formula:

$$S = Q N_A A \cdot 100\% \quad (2)$$

where *S* (%), *Q* (mol/m<sup>2</sup>), *N<sub>A</sub>* (6.022 × 10<sup>23</sup> mol<sup>-1</sup>), and *A* (m<sup>2</sup>) are surface coverage, adsorption density, Avogadro constant, and cross-sectional area of the silane molecule, respectively. Maximum coverage and coverage at 0.06 M silane concentrations are shown in Table 2.



**Figure 3.** Adsorption isotherms of HOS-C6, HOS-C10, HOS-C18, FOS-C3F3, FOS-C10F17, and KMB silanes on TiO<sub>2</sub> nanoparticles.

**Table 2.** Silane Adsorption Parameters on TiO<sub>2</sub> Nanoparticles and the Interfacial Resistance of TiO<sub>2</sub> Electrode without Dyeing

silane modifier	$Q_m$ (mol/m <sup>2</sup> )	$K_{ad}$ (dm <sup>3</sup> /mol)	$Q_{0.06}$ (mol/m <sup>2</sup> )	maximum surface coverage(%)	surface coverage at 0.06 mol/dm <sup>3</sup> (%)	$R_{rc}$ ( $\Omega$ )
nonmodified						150
FOS-C3F3	$4.9 \times 10^{-6}$	$7.5 \times 10^4$	$4.0 \times 10^{-6}$	24	20	215
HOS-C6	$7.6 \times 10^{-6}$	$16 \times 10^4$	$6.1 \times 10^{-6}$	38	30	310
HOS-C10	$5.5 \times 10^{-6}$	$13 \times 10^4$	$5.3 \times 10^{-6}$	27	26	200
HOS-C18	$3.5 \times 10^{-6}$	$4.6 \times 10^4$	$2.7 \times 10^{-6}$	17	13	700
FOS-C10F17	$17 \times 10^{-6}$	$2.6 \times 10^4$	$10 \times 10^{-6}$	85	50	1400
KMB	$4.9 \times 10^{-6}$	$14.2 \times 10^4$	$4.4 \times 10^{-6}$	24	22	60

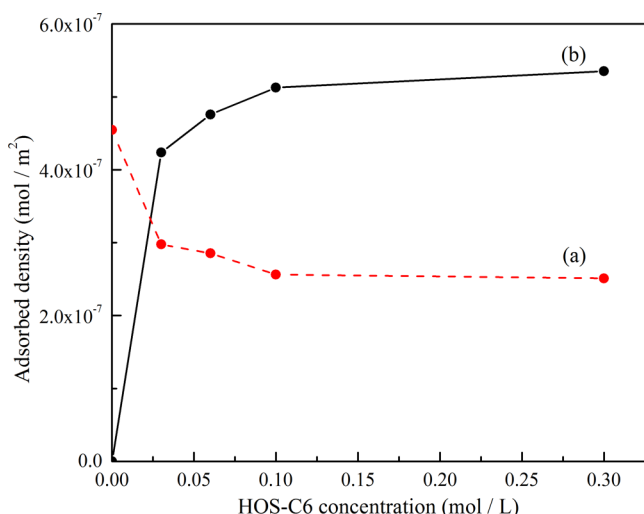
$K_{ad}$ ,  $Q_m$ ,  $Q_{0.06}$ , and surface coverage for SAMs of hydrocarbon silane decrease with increasing alkyl chain length (HOS-C6 > HOS-C10 > HOS-C18). Because, as the length of alkyl chain is increased, the interchain interactions become dominated by alkyl–alkyl interaction which leads to increase steric hindrance between silane molecules, namely, compared to long-chained silane, short-chained silane could allow more silane molecules to approach the exposed surface. FOS-C3F3 and FOS-C10F17 behave in different ways compared with hydrocarbon silanes. Surface coverage,  $Q_m$ , and  $Q_{0.06}$  increased with increasing fluorocarbon chain length and  $K_{ad}$  decreased. The highest maximum surface coverage of 85% is given with FOS-C10F17. The surface of closely packed fluorocarbon groups has the lowest possible surface energy.<sup>21</sup> The low polarizability and high-ionization potential of the C–F bond due to electro-negativity of fluorine show weak intermolecular attractive forces compared with hydrocarbon. It results in closely packed SAM with FOS-C10F17. We think that the silane adsorption would also be affected by TiO<sub>2</sub> nanoparticle surface properties, and the different facets on TiO<sub>2</sub> nanocrystals would show different silane adsorption abilities; due to their different surface energies and different Ti–OH densities on the surfaces.

Electrochemical impedance spectroscopy (EIS) is a powerful, versatile technique to study the electron transfer and the back electron transfer in DSSC system.<sup>22,23</sup> EIS was performed to investigate the effect of silanization on the back electron transfer process. The back electron transfer resistance ( $R_{rc}$ ) on TiO<sub>2</sub>/electrolyte interface (interfacial resistance) was estimated using Nyquist plot (see Figure S3 in the Supporting Information.) obtained from electrochemical impedance spectroscopy and a DSSC equivalent circuit as shown in Figure S4 in the Supporting Information.

The results are summarized in Table 2. With the exception of KMB, the value of  $R_{rc}$  increase after silanization, suggesting that they have a potential to reduce the back electron transfer. Among the DSSC cells modified with hydrocarbon silanes, HOS-C6, HOS-C10, and HOS-C18, HOS-C18 which has the longest alkyl chain shows the lowest coverage and the largest  $R_{rc}$  value. The results suggest that the  $R_{rc}$  value is dependent, not only on the coverage on TiO<sub>2</sub> surface, but also on the alkyl-chain length. FOS-C10F17 shows the highest interfacial resistance, whereas KMB is the lowest. It is possible for unsaturated benzene group in KMB to be worked in a favorable way to electron-recombination process. Therefore, further investigation was carried out using hydrocarbon and fluorocarbon silanes.

**3.2. Adsorption of Organic Silane on Dye-Sensitized TiO<sub>2</sub> Electrodes.** HOS-C6 silane was used in the studies of silane adsorption on a dye-sensitized TiO<sub>2</sub> electrode in detail. The dye-sensitized TiO<sub>2</sub> electrode was treated in various concentrations of HOS-C6 solutions. The amount of silane and the amount of dye adsorbed by TiO<sub>2</sub> electrode were calculated using weight loss corresponding to decomposition of HOS-C6 in the temperature range of 200 to 350 °C and dye in the temperature range of 350 to 450 °C in TG-DTA curves, respectively (see Figure S5 in the Supporting Information). The results (Figure 4) illustrate that with increasing the silane concentration from 0 to 0.1 M, the amount of silane uptake increases greatly and the amount of dye uptake decreases slightly, afterward both of them become nearly constant. The results suggest that silanization causes partial dye desorption from the TiO<sub>2</sub> surface.

The DSSCs were fabricated using dyed/silanized TiO<sub>2</sub> electrodes and  $I$ – $V$  performances were measured. To understand the silane modification effect on the DSSC performance,



**Figure 4.** Adsorbed densities of (a) N719 dye and (b) silane at different HOS-C6 concentrations.

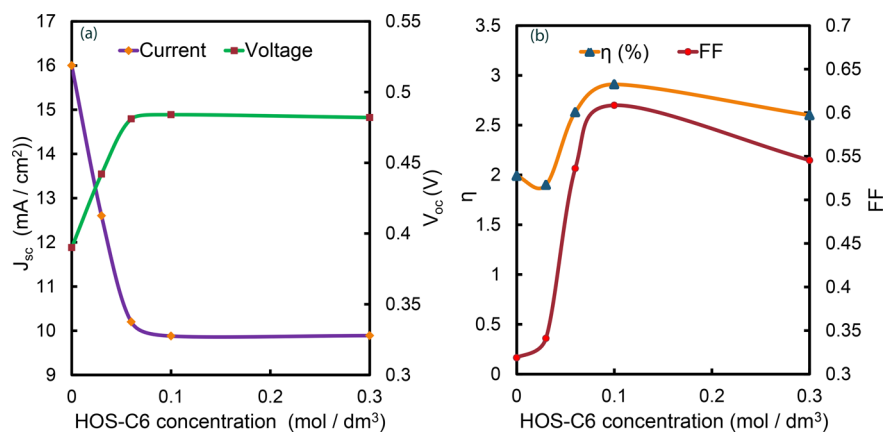
0.1 M LiI and 1.0 M I<sub>2</sub> in a mixed solvent of acetonitrile and valeronitrile (85%:15% volume ratio) was used as the electrolyte without any additives. The obtained results are shown in Figure 5. As shown in Figure 5a, open-circuit photovoltage ( $V_{oc}$ ) has increased from 0.38 to 0.48 V while increasing HOS-C6 concentration 0 to 0.06 M and then it has become constant. On the other hand, the current density ( $J_{sc}$ ) is reduced from 16 mA/cm<sup>2</sup> to 10 mA/cm<sup>2</sup>. When the silane concentration is increased, the fill factor (FF) also increases from 31% without silanization to 60% at 0.1 M silane and then continues to decrease slightly. Efficiency ( $\eta$ ) has also followed the similar pattern to the FF (Figure 5b). The improvements in  $V_{oc}$  and FF can be attributed to increase in the amount of silane adsorbed on the TiO<sub>2</sub> surface which insulates the surface from the electrolyte solution as shown in Figure 6. The silane molecules and dye molecules adsorbed on the TiO<sub>2</sub> surface block the back electron transfer from the TiO<sub>2</sub> surface to I<sub>3</sub><sup>-</sup> in the electrolyte solution. We think that one reason for the little decrease in FF at high silane concentration may be attributed to the dye desorption because the adsorbed dye molecules can also block the back electron transfer similar to silane. Under the high silane concentration conditions, the adsorption of silane molecules on the dye-uncovered area has reached almost

saturation, and then further adsorption of silane will accompany desorption of dye. The maximum coverage by both of silane and dye may reach at around 0.1 mol/dm<sup>3</sup> where shows the largest FF value. It has been reported that the FF is also affected by other factors,<sup>23</sup> these factors maybe affect the FF value. For example, a physical adsorption of silane onto the dye molecules could inhibit the dye regeneration reaction,<sup>12</sup> which will also reduce the FF value. The silane adsorption onto the dye molecules would increase with increasing the silane concentration.

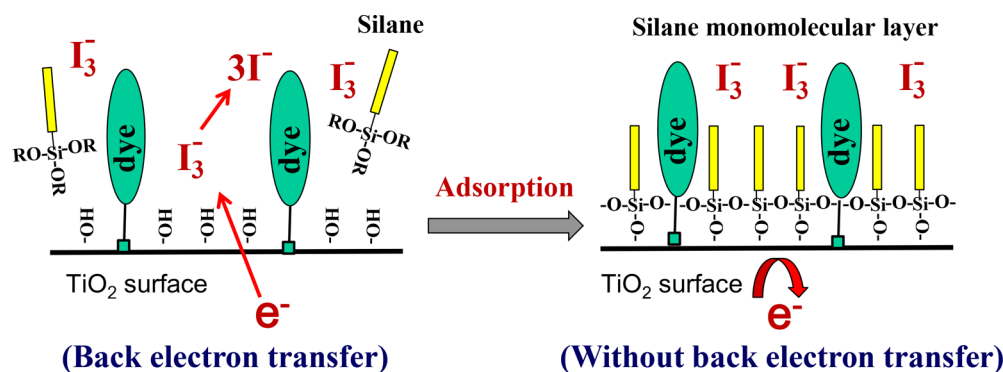
At 0.06 M silane concentration, 37% of dye is desorbed and  $J_{sc}$  is reduced by 36%. It suggests that the main reason for the reduction of  $J_{sc}$  is the dye desorption during silanization. This result is also consistent with that the silane adsorption does not adversely affect electron injection yields from dye to TiO<sub>2</sub>.<sup>14</sup> Therefore, to get optimum performance, the silane should form bonds only on the uncovered area of the TiO<sub>2</sub> electrode without damaging or removing the dye as demonstrated in Figure 6. The back electron transfer under illumination was qualitatively examined by dark-current density. As shown in dark  $I$ - $V$  curves (Figure 7), the presence of silane had a significant effect on dark-current density. Dark-current density has reduced significantly after silanization. This is consistent with the dependency of  $V_{oc}$  and FF on HOS-C6 concentration.

From these results, it was identified that dye adsorption and silane adsorption should be controlled to get high performance. In other words, silane concentration is one of the critical parameters to obtain optimum performance. It should be maintained the silane concentration to make larger surface coverage without damaging or removing dye molecules. In this study, maximum  $V_{oc}$  as well as relatively high FF and  $\eta$  were observed around 0.06 M HOS-C6 concentrations. Therefore, 0.06 M was selected for the next steps.

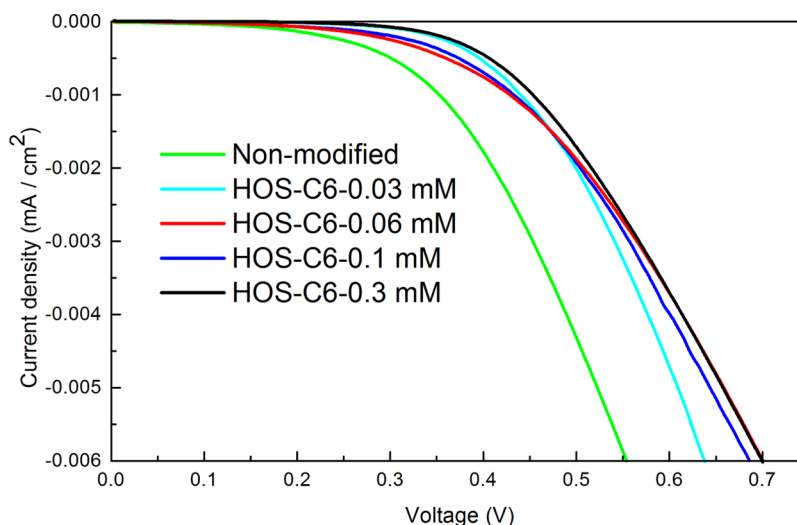
**3.3. Effect of Silanization-Process Sequence.** Although  $V_{oc}$ , FF, and  $\eta$  were improved by silanization,  $J_{sc}$  was decreased as a result of desorption of adsorbed dye after silanization. In this section, our approach is focused to improve  $J_{sc}$  by changing the TiO<sub>2</sub> electrode treatment sequence of dye and silane. The sequential adsorption of dye has been the focus of many investigations. In this process, adsorption of adsorbate can be controlled by simply changing adsorption time, concentration, and sequence of adsorption. Sequential adsorption of different type of dyes<sup>24-26</sup> and sequential adsorption of dyes and coadsorbent<sup>27</sup> in DSSC fabrication have been reported.



**Figure 5.** Dependences of DSSCs cell parameters on HOS-C6 concentration: (a) current density ( $J_{sc}$ ) and voltage ( $V_{oc}$ ); (b) fill factor (FF) and efficiency ( $\eta$ ).



**Figure 6.** Schematic representation of self-assembled silane monomolecular layer formation on  $\text{TiO}_2$  surface and blocking the back electron transfer.



**Figure 7.** Dark  $I$ - $V$  characteristics of DSSCs fabricated using  $\text{TiO}_2$  electrodes treated with different HOS-C6 concentrations.

However, the studies on the sequential adsorption of silane and dye have not yet been reported.

To improve  $J_{sc}$  by controlling the amount of silane and dye adsorbed, five types of treatment-process sequences were employed as follows: (1) dyeing without silanizing (dye), (2) dyeing followed by silanizing (dye/HOS), (3) silanizing followed by dyeing (HOS/dye), (4) simultaneous dyeing and silanizing (HOS+dye), and (5) dyeing followed by simultaneous dyeing and silanizing (dye/(HOS+dye)). HOS-C6 silane was used for silanizing. The effect of the treatment-process sequences on  $I$ - $V$  performances were investigated (see Figure S6 in the Supporting Information) and the results are summarized in Table 3. To more clearly examine the effect of a silane monolayer, we used a solution of 0.1 M LiI and 1.0 M  $\text{I}_2$  in a mixed solvent of acetonitrile and valeronitrile (85%:15% volume ratio) as the electrolyte without any additives.

The cell parameters increase in order of (HOS+dye) < dye/HOS < HOS/dye < dye < dye/(HOS+dye) for  $J_{sc}$ , dye < HOS/dye < dye/(HOS+dye) < dye/HOS < (HOS+dye) for  $V_{oc}$ , (HOS+dye) < dye < HOS/dye < dye/(HOS+dye) < dye/HOS for FF, and (HOS+dye) < dye < HOS/dye < dye/HOS = dye/(HOS+dye) for  $\eta$ , respectively. These variations in  $I$ - $V$  characteristics could be attributed to the variation in surface coverage by dye and silane as described in the previous section.

Furthermore, EIS analyses were performed to understand the effects of these processes on the back electron transfer at interfaces. Estimated interfacial resistances and electron lifetime

**Table 3.**  $I$ - $V$  Characteristics and EIS (Electrochemical Impedance Spectroscopic) Characteristics of DSSCs Fabricated Using Different Silanizing and Dyeing Treatment Sequences

process	$J_{sc}$ (mA/cm <sup>2</sup> )	$V_{oc}$ (V)	FF (%)	$\eta$ (%)	$R_s$ ( $\Omega$ )	$R_{pt}$ ( $\Omega$ )	$R_{rc}$ ( $\Omega$ )	$\tau$ (s)
dye	15.9	0.383	30	1.8	8	3	28	0.18
dye/HOS	12.2	0.444	47	2.6	6	3	100	0.36
HOS/dye	14.4	0.414	34	2.0	9	3	50	0.20
(HOS+dye)	11.6	0.453	29	1.5	5	3	79	0.63
dye/(HOS+dye)	16.5	0.442	36	2.6	7	3	140	0.71

in  $\text{TiO}_2$  electrode were made using impedance spectra (see Figure S7 in the Supporting Information) and are summarized in Table 3. The electron lifetime ( $\tau$ ) can be calculated using following formula

$$\tau = R_{rc} C_{\mu} = 1/\omega_{rec} \quad (3)$$

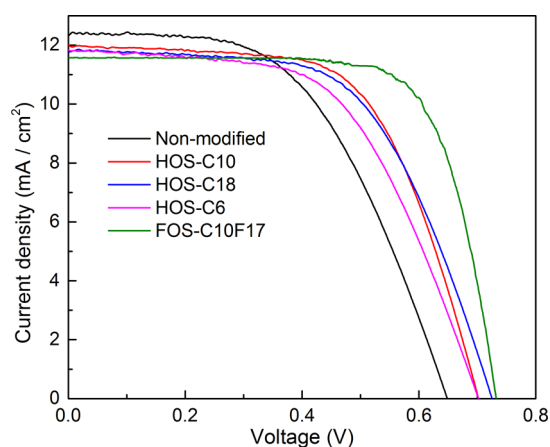
where  $R_{rc}$  is the interfacial resistance on  $\text{TiO}_2$ /electrolyte interface,  $C_{\mu}$  is the chemical capacitance, and the  $\omega_{rec}$  is the angular frequency.

Electron-transfer resistance from the working electrode to the counter electrode ( $R_s$ ) and electron-transfer resistance from the counter Pt-electrode to the electrolyte solution ( $R_{pt}$ ) have

not significantly been affected by silanization. But the interfacial resistance between electrons in the  $\text{TiO}_2$  surface and the  $\text{I}_3^-$  in the electrolyte ( $R_{rc}$ ) has been effectively improved in the every silanization process. The determined electron lifetimes are also in good agreement with the enhancements of the interfacial resistances. Although the other three processes impaired the  $J_{sc}$ , the dye/(HOS+dye) process improved  $J_{sc}$  slightly by 3.8% compared with the process without silanization. The dye/(HOS+dye) process also enhanced the  $V_{oc}$ , FF, and  $\eta$  by 15, 20, and 44%, respectively, and it is suggested to be the best process sequence. From this result, we found that photovoltaic performances can be optimized by varying the dyeing and silanizing process sequence.

**3.4. DSSC Performance Modified with Different Type of Silanes.** To investigate the effect of the different alkyl-chain length and substitutes on photovoltaic performance, we prepared DSSCs with  $\text{TiO}_2$  electrodes, silanized using four different modifiers, HOS-C6, HOS-C10, HOS-C18, and FOS-C10F17. In this section, silane modifications were performed using optimum conditions of dye/(HOS+dye), as described in the previous section, and the electrolyte solution of 0.6 M butylmethylimidazolium iodide, 0.01 M  $\text{I}_2$ , 0.1 M LiI, 0.1 M guanidinium thiocyanate, and 0.4 M 4-tert-butylpyridine in a mixed solvent of acetonitrile and valeronitrile was used in the DSSC fabrication.

Figure 8 shows the  $I$ - $V$  characteristic curves of the DSSCs and Table 4 summarize the characteristic parameters. When



**Figure 8.**  $I$ - $V$  characteristics of DSSCs fabricated using  $\text{TiO}_2$  electrodes treated with different kinds of silane solutions.

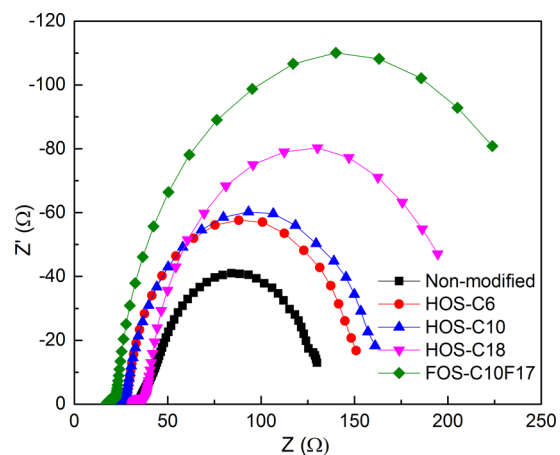
compared with unsilanized cell, every treated cell improved the  $V_{oc}$ , FF, and  $\eta$  of the DSSCs regardless of the silane structures. A small reduction of  $J_{sc}$  can be observed after silanization. It is very difficult to form the monolayer of silane only on the dye-

**Table 4.**  $I$ - $V$  Characteristics and EIS (Electrochemical Impedance Spectroscopic) Characteristics of DSSCs Fabricated Using Different Silane Modifiers

sample	$J_{sc}$ (mA/cm <sup>2</sup> )	$V_{oc}$ (V)	FF (%)	$\eta$ (%)	$R_{rc}$ ( $\Omega$ )	$\tau$ (s)
nonmodified	12.3	0.648	52.7	4.20	100	0.10
HOS-C6	11.8	0.700	56.6	4.68	125	0.16
HOS-C10	12.0	0.700	61.5	5.17	140	0.20
HOS-C18	11.7	0.730	59.2	5.06	200	0.50
FOS-C10F17	11.5	0.735	73.4	6.20	310	0.63

uncovered  $\text{TiO}_2$  surface.<sup>3</sup> The adsorption of silane molecules onto the dye molecules also may occur by a physical adsorption. Photoinduced adsorption spectroscopic studies have reported that the reduction of  $J_{sc}$  after several cycles of silanization is a result of a reduction of the dye regeneration rate.<sup>12</sup> They suggested that the silane molecules being adsorbed on the dye molecules could inhibit the dye regeneration reaction. In the present study we found that 37% of desorption of the dye after HOS-C6 silanization resulted corresponds to 36% of  $J_{sc}$  reduction. It suggested that the main reason for the reduction of  $J_{sc}$  after silanization is due to the dye desorption. We think that the silane with long-alkyl chain, such as HOS-C18, could give a steric hindrance on the oxidized dye regeneration reaction by  $\text{I}^-$ , but the effect would be small. However, silanization treatment substantially reduces the back electron transfer on the  $\text{TiO}_2$  surface, which leads to high efficiency.

Interfacial resistance and electron lifetime were determined using impedance spectra (Figure 9) to clarify  $I$ - $V$  character-



**Figure 9.** Impedance spectra of DSSCs prepared using dyeing followed by simultaneous dyeing and silanizing in different kinds of silane solutions.

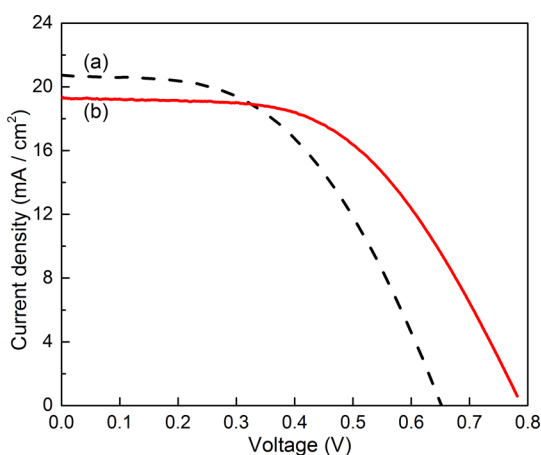
istics. The interfacial resistances ( $R_{rc}$ ) and the electron lifetimes ( $\tau$ ) derived from impedance spectra are summarized in Table 4. The  $R_{rc}$  and  $\tau$  increase in the order of nonmodified < HOS-C6 < HOS-C10 < HOS-C18 < FOS-C10F17, which is nearly consistent with the variation in the  $V_{oc}$  and FF values. The results indicate that the  $V_{oc}$  and FF are improved due to the improvement in the  $R_{rc}$  and  $\tau$ , which is a result of silanization. Increasing the alkyl-chain length of hydrocarbon silanes the  $R_{rc}$  and  $\tau$  increase and a similar trend is observed with the  $\text{TiO}_2$  electrodes without dyeing (Table 2). Although HOS-C18 showed the maximum  $V_{oc}$ , the maximum efficiency is given with HOS-C10 due to its comparatively high  $J_{sc}$  and FF values. We think that the main reason for the enhanced  $V_{oc}$  and FF are the suppression of the back electron transfer (loss of photoelectron from the  $\text{TiO}_2$  surface with electrolyte) by the silane interposed layer. It has been reported that the suppression of the back electron transfer increases  $V_{oc}$  and FF.<sup>6,8,14,28,29</sup>

Among the four modifiers, FOS-C10F17 gives the optimum  $V_{oc}$  (0.735 V), FF (73.4%) and efficiency (6.2%) that enhances the efficiency by 48% compared with unsilanized cell. These results are also in good agreement with the largest  $R_{rc}$  and  $\tau$  values, and the highest surface coverage and lowest  $K_{ad}$  obtained from adsorption isotherms (Table 2). It was found

that high surface coverage and low  $K_{ad}$  are promising conditions to form an interposed layer without damaging or removing adsorbed dye molecules.

It has been reported that except the effect of the blocking back electron transfer, the surface functionalization of the  $TiO_2$  electrode with coadsorbents can shift the  $TiO_2$  conduction band edge, which also affects  $V_{oc}$ .<sup>6,14</sup> The collective effect of a downward shift (toward positive potentials) of the band edges of  $TiO_2$  by about 100 mV and slower the rate of back electron transfer by a factor of 20-fold results to a net improvement in  $V_{oc}$  of about 20 mV has been reported with a guanidinium adsorption.<sup>9</sup> Chenodeoxycholate coadsorbent not only shifts the  $TiO_2$  band edge to negative potentials (upward shift) by about 80 mV but also accelerates the rate of back electron transfer by a factor of 5-fold, which results to a net improvement in  $V_{oc}$  of over 40 mV.<sup>30</sup> The organic silanes have a high potential to suppress back electron transfer than decylphosphonic acid, undecanoic acid, and hemin, and a 7–330-fold decrease in the rate of back electron transfer can be achieved with the silane modifications.<sup>14</sup> Therefore, the  $V_{oc}$  enhancement in the present study could be ascribed mainly due to the suppression of back electron transfer by the silane adsorptions.

To fabricate a high-performance DSSC, we silanized an optimized  $TiO_2$  electrode with scattering layer using FOS-C10F17 and the optimum conditions,  $TiO_2$  film thickness of about 14  $\mu m$ , silane treatment time of 2h, silane concentration of 0.06 mM, and silanization process sequence of (dye/(HOS+dye)), which were found in this study were employed. The  $I-V$  curves of DSSCs with silanization and without silanization are shown in Figure 10. Although the  $J_{sc}$  slightly decreases from



**Figure 10.**  $I-V$  characteristics of DSSCs fabricated using optimized  $TiO_2$  electrodes (a) without silanization and (b) with FOS-C10F17 silanization.

21.0 to 18.7  $mA/cm^2$ ,  $V_{oc}$  and FF are greatly improved from 0.65 V and 50% to 0.79 V and 54%, respectively, after silanization. Compared with DSSC without silanization (6.7%), DSSC with silanization (8.2%) showed 22%  $\eta$  enhancement. The result of this study also suggests that the {010}-faceted anatase  $TiO_2$  nanoparticles, together with the FOS-C10F17 silane surface modification technique, are promising for the fabrication of the high performance DSSCs.

#### 4. CONCLUSION

Silane adsorption on the surface of  $TiO_2$  is dependent on the alkyl-chain length and substitutes of the silane. The  $Q_m$ ,  $K_{ad}$ ,

and maximum surface coverage on  $TiO_2$  surface decrease with increasing alkyl-chain length of hydrocarbon silanes. FOS-C10F17 fluorocarbon silane gives the highest surface coverage of the silanes included in this study. The silane adsorption can enhance the  $R_{rc}$  and  $\tau$  of the  $TiO_2$  electrode, which leads to improved values for  $V_{oc}$  and FF of DSSC. This effect increases with increasing alkyl-chain length of hydrocarbon silanes, and FOS-C10F17 shows the highest effect. The silanizing causes the dye desorption from the  $TiO_2$  electrode, while the dye desorption can be suppressed by using the mixed solution of silane and dye in the silanizing process. The high performance DSSC is achieved using optimized silanizing conditions of FOS-C10F17, which yields a 22% improvement in efficiency,  $\eta$ .

#### ■ ASSOCIATED CONTENT

##### Supporting Information

TG-DTA curves of HOS-C6 adsorbed  $TiO_2$ , variation of intensity ratio of Si-O/Ti-O vibration band with adsorption time, Nyquist plots of silanized  $TiO_2$  electrodes, equivalent circuit diagram and model used to fit the impedance data, TG-DTA curves of HOS-C6 silanized and dyed  $TiO_2$  nanoparticles,  $I-V$  characteristics of DSSCs fabricated using  $TiO_2$  electrodes treated by different treatment sequences, and impedance spectra of the DSSCs fabricated using  $TiO_2$  electrodes modified with different treatment-process sequences. This material is available free of charge via the Internet at <http://pubs.acs.org>.

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##### Author Contributions

<sup>‡</sup>The manuscript was written through contributions of all authors. All authors have given approval to the final version of the manuscript. These authors contributed equally.

##### Notes

The authors declare no competing financial interest.

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